# Effect of Surface Quality on Pitting Corrosion Behavior of Aluminum Alloy 2A12

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Abstract. Surface quality is one of the most important factors of corrosion damage in aircraft structure. In this paper, quantitative method was used for surface quality, the surface roughness was obtained by scanning the specimens with a scanner. Accelerated corrosion experiments of aluminum alloy 2A12 were performed in the laboratory to analyze the effect of surface quality on the occurrence and development of pitting corrosion. The results of the experiments show that in the initial stage of the corrosion, a rough surface is more susceptible to pitting corrosion and have a larger corrosion velocity. The corrosion velocity increases with time. With the extension of time, the effect of surface quality on the corrosion behavior tends to disappear.

#### 1. Introduction

Aluminum alloy 2A12 is a high-strength aluminum-copper alloy, which has the advantages of low density, high specific strength, and good weldability. It is widely used in aircraft and other aerospace products. During the flight of an aircraft, corrosion may occur due to the corrosive effect of Cl<sup>-</sup> in the atmosphere. Among various types of corrosion, pitting corrosion is a more harmful one, and it is difficult to predict the occurrence of a pitting corrosion or to prevent it. Pitting failures may lead to sudden fractures of metal components and catastrophic accidents. Therefore, research on pitting corrosion behavior of aluminum alloys has attracted much attention. Tian-hong Zhang [1] studied the behavior of pitting corrosion of hard aluminum alloy LY12CZ, it is believed that the nitrate solution can accelerate the repair of the oxide film to improve the corrosion resistance of the material. Zhu [2] pointed out that the acidic environment in corrosion pits caused the metal on the inner wall of corrosion pits to remain in an active state, so that the anodic dissolution continued, which accelerated the enlargement of the size of corrosion pits and the deepening of pit depth. Jiang [3] studied the galvanic corrosion caused by the contact between aluminum alloy and titanium alloy, analyzed the effect of corrosion on mechanical properties and fracture mechanism of the aluminum alloy. Yang [4] studied the corrosion characteristics of aluminum alloy 5083 in seawater through experiments and found that when the cathode potential is negative to -1.15V, hydrogen evolution occurs and the

aluminum alloy dissolves. Li et al. [5] found that the relationship between mass loss due to corrosion and corrosion time shows a law of power function. Jun-guang He et al. [6] compared the pitting behavior of pure aluminum and aluminum alloys and found that the alloy can improve the corrosion morphology. Liu et al. [7] pointed out that  $CI^-$  is the active factor that induces pitting corrosion of aluminum alloys, and the polar cathodic protection potential range of 5083 aluminum alloys in seawater is obtained by polarization experiments. Li [8] used the scanning vibration electrode technique (SVET) to measure in situ the tiny area of pitting corrosion of 5083 aluminum alloy in seawater to obtain the variation of surface potential gradient. Mao-fei Zhang [9] performed experimental research and numerical analysis of aluminum alloy pitting corrosion based on near-field dynamics.

Many factors will affect the pitting corrosion behavior of aluminum alloys, including the composition of the alloy, the corrosive medium, the pH of the medium and the ambient temperature. Parts of the aircraft undergo various types of processing before assembly, such as pressure processing, metal cutting, EDM, thermal processing and so on. Different processing results in different surface quality of parts. Surface roughness is one of the important characteristics of surface quality. There are many microscopic peaks and valleys on the surface of a part, and surface roughness characterizes this microscopic geometric error. When a part is in contact with some kind of specific corrosive medium, the corrosive liquid may remain in the valley, and continuous corrosion will occur, causing localized corrosion and gradual pitting. If the pits continue to develop in the depth direction below the surface of the structure, the continuity of the material will be destroyed. This will weaken the strength of the part and cause premature failure. Pits may also be the origin of stress corrosion crack. Therefore, the influence of the surface quality of parts on the occurrence and development of pitting corrosion has become a topic of great research value. Walter et al [10] studied the effect of surface roughness on metastable pitting of magnesium alloy during initiation stage. Li et al. [11] studied the effect of surface roughness on corrosion behavior of pure copper in 3.5%NaCl solution. It was found that there was a positive correlation between the roughness of specimens and the corrosion velocity. Wang [12] used electrochemical methods to study the effect of surface roughness on the early pitting behavior of 304 stainless steel. Hou and Sun [13] studied the effect of surface roughness on corrosion resistance and resistivity of high-purity silver tape. Zhang and Wang et al [14] studied the effect of surface state on corrosion and stress corrosion behavior of alloy 690TT.

In this paper, we present and discuss the results obtained from accelerated pitting corrosion test done on aluminum alloy 2A12 in order to grasp the effect of surface quality on the occurrence and development of pitting corrosion.

# 2. Experimental procedures

In a laboratory environment, accelerated corrosion experiments of aluminum alloy 2A12 were carried out with surface roughness of the test specimen as a control variable. The incubator is used to control the corrosive ambient temperature and the treated test specimen was immersed in a formulated quantitative corrosion solution. Controlling different corrosion durations, the effect of surface quality on the occurrence and development of pitting corrosion of the test specimen under different ambient temperatures and different corrosion durations was observed by using the method of potentiodynamic scanning.

#### 2.1. Materials

The high-strength aluminum alloy 2A12 was used as the material for test specimens. The chemical composition is shown in Table 1.

Element	Si	Fe	Cu	Mn	Mg	Ni	Zn	Ti	Aluminium
Percentage	≤0.5	≤0.5	3.80~ 4.90	0.30~ 0.90	1.20~ 1.80	≤0.1	≤0.3	≤0.15	Balance

**Table 1.** Chemical composition of aluminium alloy 2A12.

# 2.2. Test specimens

The test specimen was cut by wire cutting, its shape designed as a dog bone, as shown in Figure 1. After cutting, the original specimens were processed: the aluminum-clad layer and the oxide layer were removed, cleaned and dried; the specimens were divided into two groups A and B. The group A was polished with 500# water sandpaper, and the group B is gradually polished with 500#, 1000#, 1500# water sandpaper. Scan the area of the specimen to be corroded using a Shapix 3D scanner. The results show that the maximum difference between the peak and valley values of surface roughness of group A is 8.9  $\mu$  m, while that of group B is 4  $\mu$  m.





Figure 1. Shape and size of the test specimen.

Figure 2. Corrosion morphology.

## 2.3. Experimental corrosion medium

The EXCO (exfoliation corrosion) solution is formulated according to ASTM standards. The composition is shown in Table 2. The pH of the solution is about 0.4.

Composition	NaCl	KNO3	HNO3 (70%)
Content	234g/L	64.9g/L	6.3ml/L

Table 2. Composition and ratio of EXCO solution.

# 2.4. Experimental equipment

Electrochemical tests were performed using a CS 150 electrochemical workstation. The working electrode was aluminum alloy 2A12, the reference electrode was composed of a saturated calomel electrode (SCE) and a salt bridge, and the auxiliary electrode was a Pt electrode. The relative open circuit potentials were set to -0.2 V and 0.2 V for the initial and end of the scan respectively, the scan speed was 0.5 mV/s, and the sampling frequency was 5 Hz.

# 3. Results and discussion

## 3.1. Corrosion morphology analysis

The upper part of Figure 2 respectively presented the surface morphologies of test specimens from group A, which underwent corrosion in a 25°C ambient temperature, while the corrosion time was 2

hours, 4 hours and 6 hours. The lower part, respectively presented the surface morphologies of test specimens from group B which went the same corrosion environment and corrosion time. As can be seen from the figure, the corrosion of A and B test specimens is increasing with time. However, under the same conditions, the degree of corrosion in group A was more serious than that in group B. This shows that greater roughness causes severe corrosion of the surface. When the other conditions are kept unchanged, and the ambient temperature is set to 40° C and 55° C, the above experiment is repeated and the same trend is obtained. Figure 3 respectively shows the surface corrosion state of Group A and Group B specimens observed under a microscope at an corrosion time of 2hour. As can be seen from the figure, in the same size area, group A had a number of pits of varying sizes and were independent of each other, while group B had only one pit, showing that group A corrosion was more severe than group B.

## 3.2. Potentiometric scan test

Table 3 shows the self-corrosion potential, self-corrosion current density, and corrosion velocity of aluminum alloys 2A12 with 2 different surface roughness. According to Table 3, at 25°C, as the surface roughness of the test specimen decreases, the corrosion potential moves positively from -0.77786 V in group A to -0.73949V in group B,. The corrosion current density decreased from 0.002082 A/cm<sup>2</sup> in group A to 0.002003 cm<sup>2</sup> in group B, indicating that the ion concentration in the solution decreased, the conductivity decreased, and the corrosion velocity gradually decreased. This shows that the surface quality has a direct effect on the pitting corrosion of aluminum alloy 2A12, and the larger the surface roughness is, the easier the pitting corrosion is. In the process of aluminum alloy corrosion, because of the sensitivity of aluminum to the corrosive medium with chloride ions, the chloride ions will replace the hydroxides in the aluminum hydroxide sediments by "replacement" to form highly soluble corrosion product  $AlCl_3$  and falls off from the surface of the substrate. Corrosion products cannot accumulate for a long time on the surface of the substrate. With the extension of time, the pits gradually increase, and corrosion solutions accumulate in the pits, which constitutes a localized micro-electrochemical corrosion environment, which causes the metal in the pits to continuously undergo anode dissolution. The pits develop simultaneously in both depth and radial directions. Contact area between the corrosion solution and the metal become larger, and the corrosion velocity increases. As the corrosion pit expands in the radial direction, adjacent pits will be connected to each other to form larger pits, destroying the surface integrity of the material. A large number of pits develop along the depth direction, making the material within a certain depth a porous material and destroying the continuity of the material. This will inevitably cause a decrease in the mechanical properties of the material, resulting in a shortened service life and even an abrupt failure. The above experiment was repeated while maintaining the other conditions unchanged, and the ambient temperature was set to 40° C. and 55° C. The same change rule of the electrochemical parameters was obtained.

From Figure 4, it can be seen that the corrosion velocity is increasing with time. However, the corrosion velocity in group A increased almost linearly, while in group B, the corrosion velocity hardly changed within the first 4 hours of the onset of corrosion, but it increased linearly after 4 hours. The reason is that the test specimens of group B have small roughness values and have smooth surfaces and are not prone to corrosion. Therefore, within 4 hours after the corrosion occurs, the material-medium contact surface caused by the corrosion changes very little and contributes little to the increase of the corrosion velocity. With the prolongation of time, when the corrosion develops to a certain extent, pits on the surface of the test specimen increase, the original microroughness of the material gradually disappears, the effect on the corrosion velocity tend to disappear, and the material-medium contact surface begins to rapidly increase. As a result, the corrosion velocity rapidly increases. Therefore, the difference between the corrosion velocity of Groups A and B also continues to decrease.



(a) Group A.

(b) Group B.

Figure 3. Microscopic corrosion morphology.



Table 3. Electrochemical parameters of different surface quality tests specimens in EXCO solution.

Duration(t/h)	Group	Icorr (A/cm^2)	Ecorr(V)	V(mm/a)
2	А	0.002082	-0.77786	22.736
2	В	0.002003	-0.73949	21.875
4	А	0.002187	-0.74468	23.89
4	В	0.00202	-0.73881	22.01
6	А	0.002281	-0.75811	24.917
6	В	0.002268	-0.72967	24.768

# 4. Conclusions

The accelerated pitting corrosion test was performed on 2A12 aluminum alloy test specimens to observe the microscopic surface topography after corrosion, and the effect of the surface quality of the specimens on the occurrence and development of pitting corrosion was analyzed.

1. The surface quality has a significant effect on the occurrence of pitting corrosion. The corrosion potential on the surface of the specimens with a large surface roughness value is more negative, corrosion is more likely to occur, and the corrosion velocity is higher.

2. As corrosion time increases, the corrosion velocity increases continuously.

3. In the initial stage of corrosion, the surface quality has a great influence on pitting corrosion. With the extension of time, this effect gradually disappears.

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