Chalcogenide All-solid Hybrid Microstructured Optical Fiber with Flattened Normal Dispersion and High Birefringence in the mid-IR Region

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- Keywords: Chalcogenide Microstructured Optical Fibers, Chromatic Dispersion, Birefringence Fiber, Supercontinum Generation.
- Abstract: We propose in this work a new chalcogenide hybrid microstructured optical fiber with all-solid structure to realize a flattened normal chromatic dispersion profile in a broad wavelength range which is important for highly coherent and broad bandwidth supercontinuum generation in the mid-infrared window. The calculated results show that the normal dispersion regime can be obtained from 4.5 to 13 μ m with the flatness of 4 ps/km-nm. In addition, the birefringence properties of the proposed fiber are investigated. The effects of breaking the symmetry of the fiber structure and increasing the refractive index difference between the core and the cladding solid rods can make the phase birefringence reach the values of the order of 10⁻³.

1 INTRODUCTION

Mid-infrared supercontinuum (MIR SC) light sources are expected to have many potential applications, such as bio-photonic diagnostics (Verdonck, 2016), spectroscopy (Cruz, 2015) and infrared imaging (Su, 2014) due to the presence of unique molecular absorption bands in the MIR region. Among a variety of nonlinear optical fibers which have been widely studied for the generation of MIR SC, chalcogenide optical fibers have great advantages because they have wide transmission window up to around 18 μ m and high nonlinearity (Kohoutek, 2013; Zakery, 2003; Romanova, 2014). Aiming at the generation of very broad SC spectral bandwidth, a great number of studies on SC generation by pumping highly nonlinear optical fibers in the anomalous dispersion regime have been demonstrated. However, the generated SC spectra are usually incoherent and the output SC pulses have complex temporal properties (Dudley, 2006). More recently, several studies showed that highly coherent SC generation could be achieved by pumping in the normal dispersion regime to overcome the low coherence and complex temporal properties (Klimczak, 2014; Liu, 2016; Klimczak, 2016), but

this pumping scheme drastically reduced the SC spectral bandwidth. To maintain the high coherence and improve the spectral bandwidth of SC, a broad and flattened chromatic dispersion in the normal dispersion regime is important. In addition, to minimize detrimental effects such as, polarization mode dispersion, polarization noise and instability, polarization-maintaining chalcogenide fibers which have high birefringence are potential candidates (Caillaud, 2016; Ghosh, 2019). However, high birefringence properties were demonstrated by using photonic crystal fibers with very complex elliptical air-hole structures which are very difficult to fabricate (Dabas, 2011; Yue, 2007; Hui, 2018; Saha, 2019).

In our previous work (Nagasaka, 2017), it was shown that a broadband and highly coherent MIR SC can be generated in a chalcogenide double-clad fiber using femtosecond pulses to pump in the normal dispersion regime. The fiber was experimentally fabricated by using As_2Se_3 , $AsSe_2$ and As_2S_5 chalcogenide glasses. In this work, we propose a new chalcogenide all-solid hybrid microstructured optical fiber (Ch-ASHMOF) with only one ring of 6 circular solid rods in the cladding and a high-index circular core in the center to obtain

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a more flattened chromatic dispersion in a broad wavelength range of the normal dispersion regime. Due to the absence of complex air-hole structures, the fiber fabrication is more feasible than using complex elliptical air-hole structures. The results show that flattened chromatic dispersion can be obtained from 4.5 to 13 μ m with the flatness of 4 ps/km-nm. In addition, the birefringence properties of the fiber are investigated by tuning the geometrical parameters and increasing the refractive index difference between the core and cladding rods. As a result, the phase birefringence can reach the values of the order of 10⁻³.

2 MATERIAL PROPERTIES

Figure 1a shows the refractive index profiles of As₂Se₃, AsSe₂ and As₂S₅ chalcogenide glasses. They were measured by using ellipsometry method. The refractive index difference between As2Se3 and As₂S₅ is as large as 0.6 whereas refractive indices of As₂Se₃ and AsSe₂ are close. Based on the measured refractive index data in Fig. 1a, the material dispersions of those glasses were calculated and are shown in Fig. 1b. The zero-dispersion wavelengths of As₂Se₃, AsSe₂ and As₂S₅ glasses were 6.5, 7.2 and 5.2 µm, respectively. The material dispersion of As₂S₅ rapidly increases beyond 5 µm but material dispersions of As₂Se₃ and AsSe₂ gradually increase beyond 6 µm. In Fig. 1c, transmission spectra of As₂Se₃, AsSe₂ and As₂S₅ glasses are shown. The transmission spectra were measured using a UV/VIS/NIR spectrometer (Perkin Elmer, Lambda 900) and an FTIR spectrometer (Perkin Elmer, Spectrum 100) with 1 mm-thick samples. As₂Se₃ and AsSe₂ glasses are still transparent (higher than 40%) beyond 13 µm, whereas As_2S_5 glass is not transparent over 13 μ m. An absorption band above 9 µm of As₂S₅ glass is caused by the As-S bonds. Absorption bands from 14 to 16 µm and above 17 µm of As₂Se₃ and AsSe₂ glasses are caused by the As-Se bonds. An absorption peak at 12 µm which is caused by the Se-OH bonds is found only for AsSe₂.

3 FIBER DESIGN

3.1 Flattened Chromatic Dispersion

In our previous work (Nagasaka, 2017), we demonstrated a chalcogenide double cladding fiber

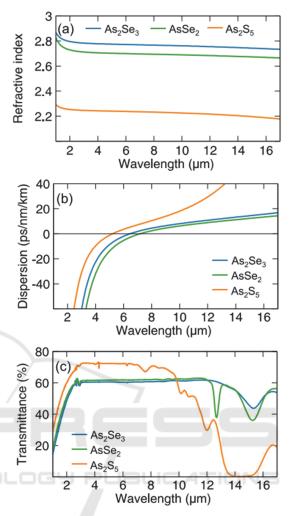


Figure 1: (a) Refractive index, (b) material dispersion and (c) transmission properties of As₂Se₃, AsSe₂ and As₂S₅ chalcogenide glasses, respectively.

(Ch-DCF) whose core, first cladding and second cladding layers were made of the As₂Se₃, AsSe₂ and As₂S₅ glasses, respectively, The diameters of the core and first cladding were $D_{core} = 7.8 \ \mu m$ and $D_{clad1} = 30.7 \ \mu m$, respectively. Its schematic cross-section image is shown in Fig. 2 and its calculated chromatic dispersion profile is shown in Fig. 3. A normal dispersion regime is obtained from 4.5 to 12.5 \ \mu m with a minimum of -12.5 ps/km-nm at 9.4 \ \mu m and anomalous chromatic dispersion is realized beyond 12.5 \ \mu m. Using 2.8-cm-long section of this fiber, we demonstrated SC generation extending from 2 to 14 \ \mu m at the -40 dB level by using the pump wavelength of 10 \ \mu m and the coupled pump peak power of 1.3 MW (Nagasaka, 2017).

To make the normal dispersion profile more flattened, a Ch-ASHMOF with an As_2Se_3 core, $AsSe_2$ cladding and a ring of 6 As_2S_5 rods is

proposed in this work. The diameters of the core, solid rods and pitch are $D_{core}=8.0$, d=1.9 and $\Lambda=13$ µm, respectively, as shown in Fig. 2. Using this Ch-ASHMOF, a normal dispersion regime whose flatness is approximately 4 ps/km-nm can be obtained from 4.5 to 13 µm as calculated and shown in Fig. 3.

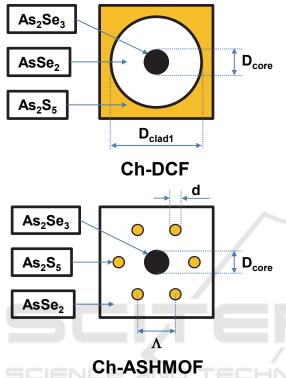


Figure 2: Schematic cross-sectional images of chalcogenide double-clad fiber (Ch-DCF) and chalcogenide all-solid hybrid microstructured optical fiber (Ch-ASHMOF).

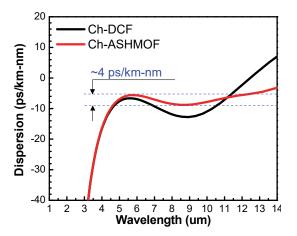


Figure 3: Calculated chromatic dispersion profiles of the Ch-DCF and Ch-ASHMOF which are depicted in Fig. 2. The calculations of chromatic dispersion profile were performed by using a commercial software (Mode Solution) from the Lumerical company.

3.2 **Birefringence Properties**

The phase birefringence $B(\lambda)$ is defined as the difference between n_x and n_y which are the real part of the effective refractive indices of x-polarized fundamental modes HE₁₁ (slow axis) and y-polarized fundamental modes HE₁₁ (fast axis) (Caillaud, 2016; Ghosh, 2019). It is calculated by using Eq. 1

$$B(\lambda) = \left| \mathbf{n}_{\mathbf{x}}(\lambda) - \mathbf{n}_{\mathbf{y}}(\lambda) \right| \tag{1}$$

Therefore, the key point of realizing the birefringence is to destroy the symmetry of fiber structure and increase the effective index difference between the two orthogonal polarization modes (Dabas, 2011). For microstructured optical fibers, high birefringence can be achieved by designing an asymmetric arrangement of air holes, not only by altering the air holes size near the core area (Caillaud, 2016; Ortigosa, 2000; Ju, 2003), but also by changing the shape of air holes (Yue, 2007), or by introducing mechanical stress as demonstrated in (Schreiber, 2005). In our work, the symmetry of the solid-rod structure is broken by modifying the properties of the L and R rods which take the core as a reflecting point as shown in Fig. 4.

Figure 4 shows the geometrical changes in the cross-sectional structures of the Ch-ASHMOFs from S0 to S1, S2a and S3a. In fact, the changes occur only for the L and R rods whereas other components keep constant. In S1 and S3a, the diameters of the L and R rods increase from 1.9 to 4 and 8 μ m, respectively, as compared to that in S0. Furthermore, in S2a, the distances from the core to the centers of

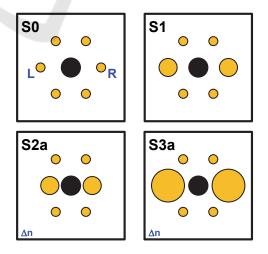


Figure 4: Schematic cross-sectional images of Ch-ASHMOFs (S0, S1, S2a and S3a). The two As_2S_5 (L and R) rods in these structures were modified to study their effects on the phase birefringence of the fiber.

the L and R rods become shorter and the diameters of the L and R rods are 4 μ m.

The calculated wavelength dependence of the phase birefringence profiles corresponding to S0, S1, S2a and S3a fiber structures are shown in Fig. 5. It can be noticed that the phase birefringence increases significantly as depicted by the red line (S2a) when the L and R rods get closer to the core because the symmetry of the fiber structure is strongly destroyed. Besides, the phase birefringence can be very high as shown by the blue line (S3a) when the diameters of the L and R rods become very different from the other As_2S_5 rods. As a result, the phase birefringence can reach 3.6×10^{-3} at $12 \,\mu$ m.

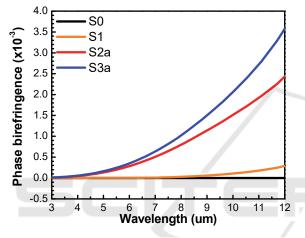


Figure 5: Calculated phase birefringence of Ch-ASHMOFs (S0, S1, S2a and S3a) whose cross-section structures were depicted in Fig. 4, respectively.

To further increase the phase birefringence, we investigate the effect of increasing the refractive index difference (Δn) between the core and the solid rods in the cladding. Figure 6 illustrates the schematic cross-sectional images of S2a, S2b and S2c structures. These structures have an identical geometry, but the refractive index of the solid rods in the cladding is assumed to be different from that of S2a, S2b and S2c. The small-grid pattern in S2b indicates that the refractive index of the solid rods in the cladding of S2b decreases by 0.05 as compared to that of the As_2S_5 in S2a. The large-grid pattern in S2c indicates that the refractive index of the solid rods in the cladding of S2c decreases by 0.1 as compared to that of the As₂S₅ in S2a. In other words, the Δn increases by 0.05 and 0.1 in S2b and S2c as compared to S2a, respectively. The calculated wavelength dependence of the phase birefringence profiles corresponding to S2a, S2b and S2c fiber structures are shown in Fig. 7 as the red solid line, dashed line and dotted line, respectively. It is

obvious that the phase birefringence increases when Δn increases, especially at wavelengths longer than 7 μm .

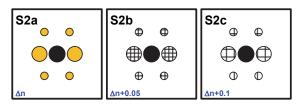


Figure 6: Schematic cross-sectional images of Ch-ASHMOFs (S2a, S2b and S2c). The refractive indices of 6 rods in the cladding of the S2b and S2c structures were assumed to decrease by 0.05 and 0.1, respectively, as compared to the measured refractive index of As_2S_5 glass which was used in the S2a structure.

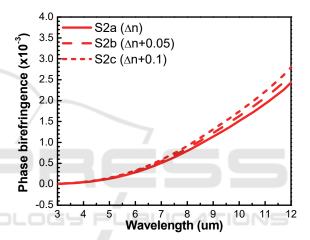


Figure 7: Calculated phase birefringence of Ch-ASHMOFs (S2a, S2b and S2c) whose cross-section structures were depicted in Fig. 6.

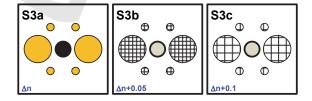


Figure 8: Schematic cross-sectional images of Ch-ASHMOFs (S3a, S3b and S3c). The refractive indices of 6 rods in the cladding of the S3b and S3c structures were assumed to decrease by 0.05 and 0.1, respectively, as compared to the measured refractive index of As₂S₅ glass which was used in the S3a structure.

Similarly, the effect of increasing Δn was investigated for the S3a fiber structure. The change in refractive index of the rods in the cladding is indicated by the small and large grid patterns in Fig. 8. The calculated wavelength dependence of

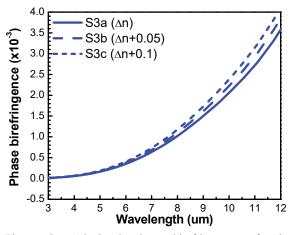


Figure 9: Calculated phase birefringence of Ch-ASHMOFs (S3a, S3b and S3c) whose cross-section structures were depicted in Fig. 8.

the phase birefringence profiles corresponding to the S3a, S3b and S3c fiber structures are shown in Fig. 9 as the blue solid line, dashed line and dotted line, respectively. As can be seen, the phase birefringence can be larger than 4.0×10^{-3} in case of the S3c structure.

4 CONCLUSIONS

In this work, a new Ch-ASHMOF made of As₂Se₃, AsSe₂ and As₂S₅ glasses with all-solid structure was proposed to realize a flattened chromatic dispersion in a broad wavelength range of the normal dispersion regime. The calculated results show that the normal dispersion regime can be obtained from 4.5 to 13 µm with the flatness of 4 ps/km-nm. In addition, the birefringence properties of the fiber were investigated by changing the diameter of two solid rods in the cladding to break the symmetry of the fiber structure and by increasing the refractive index difference between the core and the rods in the cladding. The combination of these effects can make the phase birefringence reach the values of the order of 10⁻³. Due to the feasible fabrication process, flattened chromatic dispersion and the ability of controlling phase birefringence, the proposed Ch-ASHMOF can be a good candidate to realize MIR SC generation with broad bandwidth and high coherence for many potential applications.

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